

NATIONAL ADVISORY COMMITTEE FOR AERONAUTICS

TECHNICAL NOTE

No. 1765

OFFICE OF NAVAL RESEARCH AND NACA METALLURGICAL

INVESTIGATION OF A LARGE FORGED DISC

OF S-816 ALLOY

By

Howard C. Cross Battelle Memorial Institute

and

J. W. Freeman University of Michigan



Washington February 1949

319.75 191



NATIONAL ADVISORY COMMITTEE FOR AERONAUTICS

TECHNICAL NOTE NO. 1765

OFFICE OF NAVAL RESEARCH AND NACA METALLURGICAL

INVESTIGATION OF A LARGE FORGED DISC

OF S-816 ALLOY

By Howard C. Cross and J. W. Freeman

SUMMARY

This report is one of a series in a cooperative research investigation undertaken to ascertain the properties of the better wrought heat—resisting alloys in the form of large discs required for gas—turbine rotors.

The properties of large discs of S-816 alloy have been determined for both the as-forged and aged condition (disc NR-76B-F) and the heat-treated and aged condition (disc NR-76B-Q) at room temperature and, by means of stress-rupture and creep tests for time periods up to about 2000 hours, at 1200°, 1350°, and 1500° F. Short-time tensile test, impact test, and time-deformation characteristics are included.

INTRODUCTION

This report presents the results of a study of the room-temperature, 1200°, 1350°, and 1500° F properties of a large forged disc of S-816 alloy. One-half of the disc was tested as-forged and aged, and the other half was heat-treated and aged before testing. The halves are referred to herein as two discs, NR-76B-F (forged and aged) and NR-76B-Q (heat-treated and aged).

The primary purpose of this study was to determine the level of properties exhibited by this alloy in the form of large forgings of the type required for rotor wheels of gas turbines and to determine the relative properties of such forgings as—forged and aged and as heat—treated and aged. The results obtained previously from similar investigations of 19—9DL, CSA, Timken alloy, EME, and low—carbon N—155 discs have been published as references 1 to 9. A concurrent and nearly identical investigation of a large forged disc of S—590 alloy has been published as reference 10.

This work is being carried out as part of two correlated programs of research on alloys for gas—turbine applications in progress in this country. The National Advisory Committee for Aeronautics is sponsoring work directed toward the development of improved high-temperature alloys for gas turbines used in aircraft power plants.

e and the second se

NACA TN No. 1765

A concurrent program, formerly sponsored by the National Defense Research Committee, Office of Scientific Research and Development, and now sponsored by the Office of Naval Research, Navy Department, is being directed to the development of alloys for gas—turbine applications in general and, in particular, for both ship and aircraft propulsion turbines. The work described herein was accomplished with the financial assistance of the National Advisory Committee for Aeronautics and the Office of Naval Research, Navy Department.

This report is being distributed by both the NACA and the Navy. The investigation of these discs for the NACA was conducted at the Engineering Research Institute of the University of Michigan and for the Navy at Battelle Memorial Institute.

TEST MATERIALS

The available information concerning the disc may be summarized as follows:

Manufacturer:

Allegheny-Ludlum Steel Corporation

Heat number:

41625

Chemical composition:

The chemical composition was reported by the manufacturer to be the following percentages:

<u>c</u>	<u>Mn</u>	<u>Si</u>	<u>Cr</u>	<u>N1</u>	<u>Co</u>	<u>Mo</u>	<u>w</u>	Съ	<u>F</u> Θ
0.38	0.50	0.53	19.80	20.57	42.71	3.90	4.76	3.95	2.87

Fabrication procedure:

A 12-inch-square ingot from a 5000-pound electric-arc heat was hammer cogged to $9\frac{1}{2}$ inches square from 2300° F and upset to $3\frac{5}{16}$ inches thick from 2250° F. The resulting disc was more nearly octagonal than circular and measured about 17 inches across.

The disc was cut in half. One half was left as-forged, marked NR-76-F; the other half was heated at 2300° F for $2\frac{1}{2}$ hours and water-quenched, marked NR-76-Q. Test bars from each half were then aged for 16 hours at 1400° F.



Sampling:

The code number assigned to the two halves was NR-76B. They will hereafter be referred to as the two discs, NR-76B-F and NR-76B-Q. Figures 1 and 2 show the locations of the samples cut from the halves and the code system identifying the coupons. The numbers refer to locations on the flat faces of the discs, and the letters refer to the locations through their thicknesses.

EXPERIMENTAL PROCEDURE

The investigation was designed to provide four types of information:
(1) The physical properties at room temperature, 1200°, 1350°, and 1500° F which can be expected in large forgings of the S-816 analysis; (2) the effect of heat treatment on these physical properties; (3) the variation in properties which might be present in various locations in such large forgings; and (4) the change in room-temperature properties resulting from exposure to elevated temperatures under stress for prolonged time periods.

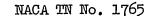
The physical-property data obtained from the halves of the large forged disc of S-816 alloy included short-time tensile properties, impact strengths, rupture test characteristics, and design curves of stress against time for total deformations of 0.1, 0.2, 0.5, and 1.0 percent at 1200°, 1350°, and 1500° F. The time — total-deformation data were obtained from time-deformation curves from both stress-rupture and creep tests.

The uniformity of the forging was checked by means of a hardness survey. Hardness, tensile, and impact tests and metallographic examination of specimens after completion of the creep tests were used to estimate the stability of the material during prolonged exposure to temperature and stress.

The testing procedures used for the short-time tensile, stress-rupture, and creep tests were in accordance with the provisions of the A.S.T.M. Recommended Practices E21-43 and E22-41.

RESULTS

The data obtained are compiled as a series of tables and figures, with the principal results from the discs NR-76B-F and NR-76B-Q summarized in figures 3 and 4. The source of the data (NACA or Navy) is indicated in the tables.





Hardness Survey

4

The Brinell hardness of material cut from the forged and aged disc NR-76B-F ranged from 258 to 311 while the range for the heat-treated and aged disc was from 255 to 280. (See figs. 5 and 6.) The hardness generally increased from the center to the rim of the discs, more so in the case of the forged and aged disc than in the case of the heat-treated and aged disc. However, the minimum value was measured about 2 inches from the center of disc Q on the center plane, while the maximum value was measured at the rim on the forged surface of disc F.

The forged and aged disc showed a higher over—all hardness than the solution—treated and aged disc, in addition to the greater hardness difference from center to rim of the forged and aged disc. The hardness variations encountered from center to rim appear to be relatively small, considering the size of the original disc from which the halves F and Q were cut and the difficulties of forging such a highly alloyed material.

Short-Time Tensile Properties

The results of the short-time tensile tests at room temperature, 1200°, 1350°, and 1500° F are shown in table T. At room temperature, 1200°, and 1350° F, the forged and aged disc NR-76B-F showed slightly higher tensile and yield strengths than the heat-treated and aged disc NR-76B-Q. At 1500° F, the situation was reversed, with the heat-treated and aged disc showing slightly higher strengths than the forged and aged disc. The variation in ductility was small, both between discs F and Q and among the room-temperature and elevated-temperature tests. However, at room temperature and at 1350° F, the average elongation for disc Q was a few percent above that for disc F. The elongations of both discs were higher at room temperature and 1350° F than at 1200° and 1500° F. It should be noted, however, that every specimen tested at 1200° F broke in the gage marks. The average elongation for all the tests on both discs was about 20 percent.

There was a tendency for the specimens from the interior of the discs to have lower strength at room temperature and, in disc Q, lower ductility.

Charpy Impact Resistance

Charpy impact resistance (V-notch) was determined on specimens from the two discs F and Q at room temperature, 1200°, 1350°, and 1500° F after holding at temperature sufficiently long to insure a uniform temperature in the specimens. The data are shown in table II and figures 3 and 4.



NACA TN No. 1765

The Charpy impact values of the discs F and Q were nearly identical at 1200° and 1500° F, but at room temperature and at 1350° F the impact strength of the forged and aged disc was higher than that of the heat-treated and aged disc. For both the discs F and Q, the impact strength at the elevated temperatures was nearly twice that at room temperature and did not change significantly from 1200° to 1500° F.

The average of the impact strengths of the specimens from the surface of the discs was about 20 percent higher than that of the interior specimens for both discs F and Q at all temperatures.

Rupture Test Characteristics

The stress-rupture data for the tests at 1200°, 1350°, and 1500° F are shown in table III, and the rupture-strength values obtained from the curves of stress against rupture time in figures 7 to 13 are shown at the bottom of table III. All stress-rupture tests were run on 1/4-inch-diameter specimens which were cut from the discs in either a radial direction or less than 45° from radial.

At 1200° F, the stress-rupture strength of the heat-treated and aged disc Q was slightly superior to that of the forged and aged disc F. The 100-hour and 1000-hour rupture strengths of disc Q were 66,000 and 52,000 psi, respectively, and of disc F 62,000 and 50,000 psi, respectively.

At 1350° F, the rupture strengths of the forged and aged disc and quenched and aged disc were nearly identical; the 100-hour and 1000-hour strengths were about 38,000 and 29,000 psi, respectively.

At 1500° F, the 100-hour and 1000-hour rupture strengths of disc Q were again slightly higher than those of disc F. The 100-hour and 1000-hour strengths of disc Q were 23,000 and 17,500 psi, respectively, and of disc F 21,000 and 13,500 psi, respectively.

Inspection of the curves of stress against rupture time (fig. 7) shows some increase in slope with increasing test temperature for disc F and hardly any change in slope with increasing test temperature for disc Q. At 1200° and 1350° F, the slopes of the stress-rupture curves are about the same for both discs. The comparative slope changes between discs at 1500° F indicate that for service at this temperature, the heat—treated and aged disc was superior to the forged and aged.

Ductilities of the stress-rupture specimens measured after fracture varied from fair to good. Elongations for the 1500° F tests were mostly only fair, ranging from 2 to 8 percent, while those for the 1200° and 1350° F tests were better, ranging from 6 to 15 percent, with one low value, 3 percent, reported at 1350° F (disc NR-76B-F).

NACA IN No. 1765

6

Time-Deformation Characteristics

A convenient method of describing the high-temperature strength of a material is by curves of stress against time required for various total deformations. Data from both stress-rupture and creep tests are used to prepare such design curves. Such information, along with the curves of stress against rupture time, gives design engineers a complete picture of the expected performance of an alloy under constant tensile stress. This information is presented in figures 8 to 13 for deformations of 0.1, 0.2, 0.5, and 1.0 percent at 1200°, 1350°, and 1500° F for time periods up to 2000 hours. Curves showing the time of transition from a minimum creep rate to the increasing rate of third-stage creep have been added so as to show when rapid elongation preceding failure starts.

The curves of stress against time for total deformation were plotted from the data in tables III to VI. The data were taken from the time—deformation curves of the stress—rupture and creep tests. The time—deformation curves for the creep tests and stress—rupture tests have not been included in this report.

Tables IV, V, and VI also show data scaled from the design curves in figures 8 to 13 and show the stresses to cause various total deformations from 0.1 to 1.0 percent in definite time periods of 1, 10, 100, 1000, and 2000 hours. For ease of comparison, these data and similar data for the S-590 discs NR-74B and for the N-155 disc NR-66D are shown together in table VII. Also included for comparison are impact and tensile data at room temperature, 1200°, 1350°, and 1500° F for these materials and residual room-temperature impact and tensile data for specimens after creep testing of the various materials (after 1000 to 2000 hr under stress at test temperature).

Creep Strengths

Many engineers are accustomed to basing designs on creep rates, especially for long periods of service. For this reason, the creep rate data have been collected from the time-deformation curves and are shown in table VIII, and the logarithmic curves of stress against creep rate for the tests at 1200°, 1350°, and 1500° F on the two discs F and Q are shown in figure 14. The creep rates used were either minimum or final rates from 1000-hour tests at 1200° F and from 2000-hour tests at 1350° and 1500° F. The creep strengths obtained from figure 14 are shown in tables IV, V, and VI and for ease of comparison are tabulated as follows:



	Temperature	Stress (psi) for creep rates of -								
Disc	- (°F)	0.0001 percent/hr	0.0000l percent/hr							
NR-76B-F	1200	28,000	a18,000							
NR-76B-Q		28,000	a16,000							
NR-76B-F	1350	20,000	13,000							
NR-76B-Q		19,000	10,500							
NR-76B-F	1500	11,000	8,500							
NR-76B-Q		13,500	7,500							

aEstimated.

NACA TN No. 1765

These creep strengths can be compared with the deformation strengths in tables IV, V, and VI. The creep strengths for a rate of 0.0001 percent per hour at 12000 F are apparently safe for use for time periods up to 10,000 hours since extrapolation of the transition-point curves (stage-two to stage-three creep rate) in figures 8 and 9 out to 10,000 hours indicates that at the stresses listed second-stage creep would still prevail.

For the tests at 1350° F, the situation is quite different. Extrapolation of the 1350° F transition curves in figures 10 and 11 shows that at the stresses for a creep rate of 0.0001 percent per hour increasing creep rates will occur in about 14,000 hours for disc F and 7000 hours for disc Q. At the lower stresses producing a creep rate of 0.00001 percent per hour, third-stage creep would not occur for considerably longer time periods.

Extrapolation of the 1500° F transition curves (figs. 12 and 13) shows that at the stresses for a creep rate of 0.0001 percent per hour. increasing creep rates will occur in 1500 hours for disc F and in 3700 hours for disc Q. Even at the stress for a creep rate of 0.00001 percent per hour, extrapolation indicates an increasing creep rate for disc F in 4000 hours and for disc Q at about 25,000 hours.

Comparison of the extrapolated rupture strengths with the creep strengths indicates that the estimated stresses for fracture in 10.000 hours are well above the 0.0001-percent-per-hour creep strengths at 1200° F. The rupture strength (10,000 hr) is only slightly higher than the creep strength (0.0001 percent/hr) at 1350° F. At 1500° F. the creep strength of disc F is above the estimated rupture strength, while for the disc Q the two are equal.

. The scatter of the creep-rate data and the change in slope downward of the curve of stress against creep rate at low stresses producing rates less than 0.0001 percent per hour make unwise the large extrapolations necessary in both creep and rupture data for comparisons of the creep rate for 0.00001 percent per hour with the 100,000-hour rupture strengths. 8

NACA IN No. 1765

In all cases, caution should be observed since extended service periods of several times the maximum test period used here (about 2000 hr) may magnify such effects as surface and structural instability, which were overshadowed by other variables during the short test periods.

Stability Characteristics

Some of the test specimens from each of the two discs were subjected to tensile, impact, and hardness tests at room temperature after creep testing at 12000, 13500, and 15000 F with the results shown in table IX.

The considerable decreases in impact strength and tensile test ductility at room temperature were the most significant changes observed.

The hardness of disc F increased somewhat after creep testing at 1200° F, but did not change after testing at 1350° F and seemed to decrease slightly after testing at 1500° F.

For the forged and aged disc F, the proportional limit and 0.1—and 0.2—percent yield strengths were slightly higher after creep testing at 1200° and 1350° F (except for the 0.1—percent yield strength after 1200° F which remained unchanged), but were lower after testing at 1500° F. The tensile strength was lower after creep testing at all three temperatures, and the 0.02—percent yield strength did not change significantly after testing at 1200° and 1350° F but decreased after testing at 1500° F.

The yield strengths of the heat-treated and aged disc Q were higher after creep testing at 1200° and 1350° F and lower after testing at 1500° F. The tensile strength was lower after creep testing at all three temperatures.

The microstructure showed less breakdown of the dendritic pattern at the center of the disc than near the rim, and the grains were coarser at the center. The structure near the rim was, however, more typical of that of the test specimens. Original microstructures of the center and rim portions of the forged and aged disc are shown at magnifications of 100X and 1000X in figure 15. The microstructure of the rim portion of the heat—treated and aged disc is also shown at 100X and 1000X in figure 15.

The grain-size range was about 5 to 7 near the rim of the forged and aged disc and about 4 to 7 at the center. The grain-size range near the rim of the heat-treated and aged disc was about 4 to 6. Thus the heat-treated and aged disc was slightly coarser grained (by one A.S.T.M. grain-size number) than the forged and aged disc.

Precipitation within the grains is not evident in the original microstructures or in those of the creep—tested specimens. Some precipitation is evident in the microstructures at 1000X of specimens 10A, Q12F, and Q12E (figs. 16 and 17), which were tested in stress

NACA IN No. 1765

rupture, the first two at 1350° and the last at 1200° F. No explanation can be offered for the apparent absence of precipitation in the photomicrographs of the original structures and of the structures of all the tested creep specimens and 1500° F stress-rupture specimens. The photomicrographs of figure 15 show the original structures at magnifications of 100X and 1000X of the forged and aged and heat-treated and aged discs. Those of figures 18 and 19 show the structures at 100X and 1000X of the two discs, F and Q, after creep testing at 1200°, 1350°, and 1500° F. The photomicrographs of figures 16 and 17 show the fractures at 100X and internal microstructures at 1000X of the specimens of the two discs after stress—rupture tests at 1200, 1350, and 1500 F. The fracture of specimen Q12E, which was tested for 1699 hours at 1200° F, appears to be transcrystalline. The fractures of the specimens tested at 13500 (FIOA and Q12F) and 1500° F (FIID and Q14D), and the fracture of specimen F9A (2618 hr at 1200° F) were all intergranular. The longer time of exposure to high temperature and lower stress may have caused specimen F9A to break with an intergranular fracture at 1200° F, while the corresponding specimen Q12E from the quenched and aged disc broke with a transcrystalline fracture. Specimens tested in stress rupture at 13500 and 15000 F (FlOA, Q12F, FllD, and Q14D) showed considerable intergranular cracking adjacent to the fracture.

DISCUSSION OF RESULTS

The tensile, stress-rupture, creep, and time-deformation data provide as nearly complete design information for the S-816 discs, NR-76B-F and NR-76B-Q, as can be obtained in the laboratory from tests under constant tensile stress.

The test data contained in this report apply only to the particular discs tested and fabricated and heat—treated in the manner indicated. Considerable experience indicates that the properties depend on the particular manufacturing process used in the production of the discs. It should not be assumed that the properties herein reported apply to discs of a similar composition produced by another fabricator or necessarily to similar discs produced by the same fabricator.



SUMMARY OF RESULTS

The principal results obtained from the 17-inch-diameter by $3\frac{5}{16}$ -inch-thick discs may be summarized as follows:

	Forged and aged disc,	Heat-treated and aged disc, NR-76B-Q
1. Brinell hardness range: On center plane at rim On center plane at center	300 270	280 260
2. Offset yield strengths:		
0.2-percent-offset yield strength, psi, at - Room temperature 1200° F 1350° F 1500° F	85,600 67,000 59,000 49,000	76,500 58,000 54,500 51,000
3. Rupture test characteristics:		
Stress, psi, to cause rupture at 1200° F in — 10 hours 100 hours 1000 hours	78,000 62,000 50,000	80,000 66,000 52,000
Stress, psi, to cause rupture at 1350° F in — 10 hours 100 hours 1000 hours	52,000 38,000 29,000	52,000 39,000 29,000
Stress, psi, to cause rupture at 1500° F in — 10 hours 100 hours 1000 hours	30,000 21,000 13,500	29,000 23,000 17,500

The elongations and reductions of area of the fractured rupture test specimens were generally satisfactory with the exception of some specimens of the forged and aged disc, NR-76B-F, tested at 1500° F. Increased rupture time did not produce a significant change in ductility.



4. Total-deformation characteristics under stress:

The data for the two discs are shown elsewhere in this report and will not be repeated here. Briefly, the heat-treated and aged disc was generally superior at 1200° F, although extrapolation of the data obtained indicated that for long times at low stresses (0.1- and 0.2-percent total deformation) the forged and aged disc might be superior. The forged and aged disc was superior at 1350° F, and at 1500° F the solution-treated and aged disc was superior except at low stresses (0.1-percent total deformation). No explanation has been found for the fact that the forged and aged disc was quite definitely superior at 1350° F, while the heat-treated and aged disc was superior at both 1200° and 1500° F.

For very long time service, the order of superiority described might be changed somewhat. This is evidenced by the fact that the creep strengths (0.0001 percent per hour and 0.00001 percent per hour) of the forged and aged disc were equal or superior to those of the heat—treated and aged disc at all three test temperatures, except for the 0.0001—percent—per—hour creep strength at 1500° F where the heat—treated and aged disc had the higher strength.

5. Uniformity:

The properties of the discs were quite uniform in view of the size of the forging and the characteristics of the alloy.

6. Stability:

The impact strength and ductility decreased after creep testing at 1200°, 1350°, and 1500° F. The ultimate-strength values from tensile tests decreased after creep testing at all three temperatures. The yield strengths of the forged and aged disc changed very little after creep testing at 1200° and 1350° F and decreased after testing at 1500° F. The yield strengths of the heat-treated and aged disc increased after creep testing at 1200° and 1350° F and decreased after creep testing at 1500° F.

Battelle Memorial Institute Columbus, Ohio

and

University of Michigan Ann Arbor, Mich. March 17, 1948



REFERENCES

- 1. Freeman, J. W., Reynolds, E. E., and White, A. E.: A Metallurgical Investigation of a Large Forged Disc of 19—9DL Alloy. NACA ACR No. 5C10, 1945.
- 2. Freeman, J. W., Reynolds, E. E., and White, A. E.: A Metallurgical Investigation of a Large Forged Disc of CSA (234-A-5) Alloy. NACA ARR No. 5H17, 1945.
- 3. Freeman, J. W., and Cross, H. C.: A Metallurgical Investigation of a Large Forged Disc of Low-Carbon N-155 Alloy. NACA ARR No. 5K20, 1945.
- 4. Cross, Howard C., and Freeman, J. W.: A Metallurgical Investigation of Large Forged Discs of Low-Carbon N-155 Alloy. NACA IN No. 1230, 1947.
- 5. Freeman, J. W., Reynolds, E. E., and White, A. E.: A Metallurgical Investigation of Five Forged Gas—Turbine Discs of Timken Alloy. NACA TN No. 1531, 1948.
- 6. Freeman, J. W., Reynolds, E. E., and White, A. E.: A Metallurgical Investigation of Two Contour-Forged Gas-Turbine Discs of 19-9DL Alloy. NACA TN No. 1532, 1948.
- 7. Reynolds, E. E., Freeman, J. W., and White, A. E.: A Metallurgical Investigation of Two Large Discs of CSA Alloy. NACA TN No. 1533, 1948.
- 8. Reynolds, E. E., Freeman, J. W., and White, A. E.: A Metallurgical Investigation of a Contour-Forged Disc of EME Alloy. NACA TN No. 1534, 1948.
- 9. Reynolds, E. E., Freeman, J. W., and White, A. E.: A Metallurgical Investigation of Two Turbosupercharger Discs of 19-9DL Alloy. NACA IN No. 1535, 1948.
- 10. Freeman, J. W., and Cross, Howard C.: NACA and Office of Naval Research Metallurgical Investigation of a Large Forged Disc of S-590 Alloy. NACA TN No. 1760, 1949.



TABLE I.- SHORT-TIME TENSILE PROPERTIES OF S-816 ALLOY DISCS IR-76B

Pulled at 0.02 in./min through yield strengths, then 0.06 in./min to rupture

Disc (a)	Specimen number	Specimen location	ltemperature	Tensile strength	Offe	et yield stre (psi)	engths	Proportional	Elongation in 2 in.	Reduction of area	
(a)	mmoer	LOGATION	(_{JL})	(psi)	0.02 percent	0.1 percent	0.2 percent	1 7	(percent)	(percent)	elasticity
NR-76B-F (forged)	bf19x bf18x bf17y bf18y	Surface Surface Interior Interior	Room Room Room Room	153,000 150,300 149,400 147,100	66,800 67,400 60,000 58,000	84,000 82,500 75,000 74,500	91,000 89,000 81,200 81,000	48,600 47,400 43,600 37,400	20.5 18.9 23.0 19.5	19.9 19.5 23.4 19.2	30.4 × 10 ⁶ 34.0 32.5 33.5
	ofax of6z	Surface Surface	1200 1200	117,500 121,750	53,000 51,800	63,000 63,500	67,000 67,000	41,500 42,000	d _{14.5} d ₁₇	14.1 15.9	26.5 25.0
	orgi orgi	Surface Interior	1350 1350	94,000 83,250	53,700 44,400	61,600 50,000	66,000 52,800	39,000 36,000	23.5 22.5	24.4 22.0	25.0 23.8
	prox prex	Surface Interior	1500 1500	57,500 60,400	35,200 40,500	44,000 48,500	46,800 51,000	27,000 31,000	16.5	19.9	24.1 24.0
MR-76B-Q (quenched)	р63Х р65Х р63Х р63Х	Surface Surface Interior Interior	Room Room Room Room	149,500 149,000 140,300 138,400	58,800 55,900 55,000 56,400	71,800 71,400 68,800 70,400	77,800 77,400 74,400 76,400	35,900 34,400 30,000 32,900	32.5 33.0 18.5 16.2	25.9 29.0 16.9 15.7	35.2 33.3 32.1 32.5
	°07 x °08 y	Surface Interior		107,750 105,500	48,500 50,000	54,300 56,700	57,000 60,000	37,500 37,500	d18	14.8 13.4	27.2 25.8
	• •	Surface Surface	1350 1350	84,000 83,000	44,500 48,500	50,000 54,200	52,600 56,800	35,000 40,000	25.5 23	22.7 21.6	25.0 24.0
	. •	Surface Interior	1500 1500	58,500 63,200	42,400 42,400	48,500 5 0,000	50,200 51,500	40,000 34,900	19.6 16.0		29.5 17.0

**Heat treatments:

NR-76B-F: As-forged and agod; 16 hr at 1400° F, air-cool.

NR-76B-Q: Heat-treated and agod; 2300° F, 2½ hr water-quenched; 16 hr at 1400° F, air-cool.

bRavy data.

CMACA data.

dBroke in gage mark,



TABLE II.- CHARPY NOTCHED-BAR IMPACT RESISTANCE AT ROOM TEMPERATURE $1200^{\rm o},~1350^{\rm o},~{\rm and}~1500^{\rm o}$ f for s-816 alloy discs NR-76B

Navy data: 0.394-in. square specimens with a 0.079-in.-deep V-notch

Disc (1)	Specimen number	Specimen location	Test temperature (°F)	Charpy impact strength (ft-lb)	Average Charpy impact strength (ft-1b)
NR-76B-F (forged)	8D 14X6 15X4 7A	Interior Interior Interior Surface	Room Room Room Room	15 27 27 22	23
	15X1 14Z1	Surface Surface	Room Room	27 36	28
	7D 1426 15x6	Interior Interior Interior	1200 1200 1200	31 47 43 43	40
	9F 14X3 15X2	Surface Surface Surface	1200 1200 1200	43 54 39	45
	14X4 14X5 15X5	Interior Interior Interior	1350 1350 1350	41 42 39	41
	14X1 14Z2 15X3	Surface Surface Surface	1350 1350 1350	39 45 61 54	53
	70 1424 1425	Interior Interior Interior	1500 1500 1500	31 50 48	43
	7F 14X2 14Z3	Surface Surface Surface	1500 1500 1500	48 34 50	ի ի
NR-76B-Q (quenched)	150 160 188	Interior Interior Interior	Room Room Room	18 16 18	17
	8x1 18F	Surface Surface	Room Room	21 24	22
	15B 17C 19D	Interior Interior Interior	1200 1200 1200	42 39 38 47	40
	15A 17F 18A	Surface Surface Surface	1200 1200 1200	47 .50 42	46
	150 160 180	Interior Interior Interior	1350 1350 1350	38 35 38	37
	8X2 16F 19A	Surface Surface Surface	1350 1350 1350	37 51 43	44
	170 180 190	Interior Interior Interior	1500 1500 1500	40 34 36	37
	15F 17A 19F	Surface Surface Surface	1500 1500 1500	51 48 50	49

Heat treatments:

NR-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool.

NR-76B-Q: Heat-treated and aged; 2300° F, 21/2 hr, water-quenched; 16 hr at 1400° F, air-cool.



NACA TN No. 1765

TABLE III.- RUPTURE TEST DATA AT 1200°, 1350°, AND 1500° F FOR S-816 ALLOY DISCS NR-76B

		Γ	Test	1	Runtiine	El onget to	n Reduction	Minimum
Disc (a)	Specimen number	Specimen location	temperature (°F)	Stress (ps1)	time (hr)	in 1 in.	of area	creep rate (percent/hr)
NR-76B-F (forged)	bf9E bf9C bf1Œ bf9B bf9A	Interior Interior Interior Interior Surface	1200 1200 1200 1200 1200	60,000 55,000 55,000 50,000 45,000	246 638 658	°8 °11 13 °10 °7	9 10.5 13.8 9.4 6.4	0.050 .034 .015 .0068 .0012
	bF9D bF10C bF10B bF10F bF10D bF10A	Interior Interior Interior Surface Interior Surface	1350 1350 1350 1350 1350 1350	60,000 40,000 35,000 35,000 30,000 27,000	212 275 565	°15 °11 °10 8 14 °3	16.2 12.1 10.2 8.6 12.5 4.7	.028 .018 .0145 .0009
	d _{F11A} d _{F11B} d _{F11E} d _{F11C} d _{F11D}	Surface Interior Interior Interior Interior	1500 1500 1500 1500 1500	25,000 21,000 18,000 16,500 13,500	81 236 362	2 3 8 2 4	2.4 5.5 5.6 .8 2.3	.0175 .0137 .010 .0015 .0014
NR-76B-Q (quenched)	poise poise poise poise	Surface Interior Interior Interior	1200 1200 1200	65,000 60,000 55,000 50,000	203 683	°7 °6 °7 °7	10.2 4.8 7.1 4,5	.025 .0075 .0026
	bolsa bolse bolse bolse	Surface Interior Interior Surface	1350 1350 1350 1350	40,000 35,000 30,000 27,500	253 868	°11 °14 10 8	14.7 9.8 9.8 10.2	.030 .0047 .0023
	dqi4A dqi4B dqi4C. dqi4D dqi4F	Surface Interior Interior Interior Surface	1500 1500 1500 1500 1500	25,000 22,000 19,500 17,500 16,000	136 443 928	7 7•5 5 4	7.0 7.8 4.7 3.2	.046 .012 .0026 .0015
			Rup	ture str	ength			
Disc		Temperati (°F)			Т		rupture in	·
NR-76B-F (forged)		1200 1350 1500	f	10 hr 75,000 52,000 30,000	100 62, 38, 21,	000	1000 hr 50,000 29,000 13,500	2000 hr 147,000 26,500 f12,000
NR-76B-Q (quenched)		1200 1350 1500	17	30,000 52,000 29,000	66, 39, 23,	000	52,000 29,000 17,500	49,000 f26,000 f16,000

^BHeat treatments:

NR-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool.
NR-76B-Q: Heat-treated and aged; 2300° F, 22 hr, water-quenched; 16 hr at 1400° F, air-cool.

bNACA data; 0.250-in.-diameter specimens with a 1-in. gage length.

Chroke in gage mark.

Navy data; 0.250-in.-diameter specimens with a 1.3-in. gage length.

The extensometer was insecure during this test and deformation readings after 50 hr were in error because of rupture of adjacent units. The test was discontinued at 527 hr. Measurement of the specimen after removal from test showed 1.0-percent elongation and 2.4-percent reduction of area.

Estimated values extrapolated from figs. 7 to 13.



TABLE IV. - DATA FOR STRESS AND TIME FOR TOTAL DEFORMATION AT 1200° F FOR 8-816 ALLOY DIRCS NR-76B MACA data

			Initial		Time (hr) fo	or total defo	rmations o	 of ~		Transition t	o third-stage creep		
Disc (a)	Specimen number	Stress (psi)	detornation	0,1 percent		0.5 percent			5 percent	Time (hr)	Deformation (percent)		
MR-76B-F (forged)	qeor qeor qeor qeor qeor qeor qeor qeor	25,000 35,000 45,000 50,000 55,000 60,000	0,087 ,133 ,196 ,210 ,250 ,365 ,300	 	875 30 1 	503 100 11 4 7	**************************************	 970 92 52 58 17	347 139 208 68	1840 625 160 400 120	3.2 7.4 5.7 7.8 7.1		
NR-76B-Q (quenched)	9 01 04 9 01 04 9 01 08 9 01 08 9 06 08 9 06 08	24,000 35,000 50,000 55,000 60,000 65,000	.276 .351 .440	25 	1480 85 	1030 14 14 03	163 85 15	415 182 44 17	1480 550 168 72	1630 450	5.4 4.0		
Dis	3	T	otal deforma	tion		8tr	ess (psi)	to bewse t	otal defor	mation in -			
(a)			(percent)		1 hr	10	hr	100 h	r	1000 hr	2000 hr		
IIR-76 (forg	B-37 e-d.)		0.1 .2 .5 1.0 Transition		45,000 61,000	38	,500 ,000 ,000 ,000	31,50 43,00 48,00 59,50	00000	24,500 33,500 38,000 48,000	°22,500 30,500 35,500 44,500		
NH-76 (queno	B—Q hed)		.1 .2 .5 1.0 Transition		°65,000 °71,000	⁰ 43	,000 ,000 ,500 ,000	°19,50 34,50 46,00 52,50 °64,00	0	25,500 36,000 43,000 52,000	023,000 033,000 040,000 48,000		
		Disc				Creep s	trength (p	si) at 100	Ohrfor (neeb refee o	· _		
		(a)			0.0001 percent/hr					0.00001 percent/hr			
	•	F (for	_		28,000				°18,000				
	NR-76E	}-Q (que	nohed)			28,000				°16,0	20		

Effect treatments:

MR-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool.

NR-76B-Q: Heat-treated and aged; 2300° F, 2½ hr, vater-quenched; 16 hr at 1400° F, air-cool.

bCroop tests; 0.505-in.-diameter specimens with a 2-in. gage length. Estimated values.

dStress-rupture tests; 0.250-in.-diameter specimens with a 1-in. gage length.



TABLE V.- DATA FOR STRESS AND TIME FOR TOTAL DEFORMATION AT 1350° F FOR 8-816 ALLOY DIROS ER-768

			Initial		Time ()	r) for total de	formations	of -		Transition	to third-stage oreep
Disc (a)	Specimen number	Strees (psi)	deformation (percent)	0.1 percent	0,2 percent	0.5 percent	1 percent	2 percent	5 percent	Time (hr)	Deformation (percent)
IR-76B-F (forged)	orios orios orios orios orios orios orios orios orios	12,000 15,000 20,000 20,000 35,000 35,000 35,000	0.057 .072 .080 .094 .113 .125 .145 .164	400 15 4 1	1075 160 110 9 1.5 1.5	1879 1879 185 10 5 10	800 31 16 40 de 5	1695 86 47 100 6	 300 144 245 20	1600 300 100 130	1,8 5,2 3,5 2,5
MR-76B-Q (quenched)	od3v od3b od50 od51 od1v pdn pdn pdn pdn	12,000 12,000 15,000 18,000 27,500 30,000 40,000	.057 .053 .078 .078 .096 .115 .126 .140	175 225 28 8 1	43000 48000 580 132 50 5 2,5	 d ₃ 800 890 58 33 42 <1	243 124 8	688 343 36 10	 845 125 38	 820 440 100 45	2.3 2.5 4.0 6.0
Dis		Tot	al deformation		Stress (psi) to cause total deformation in -						
(a)			(percent)		1 hr	10 hr		100 hr 1000		·	2000 hr
mn-7 6 (forg			0.1 .2 .5 1.0 Tremaition		24,000 33,000 39,000 43,000	17,000 27,500 33,000 37,000		13,000 22,000 28,000 31,000 35,500	^d 10,000 16,500 24,500 ^d 26,500 28,000		49,500 415,000 23,000 425,000 426,000
IR-76 (quenc			.1 .2 .5 1.0 Transition		21,000 430,000 38,500 443,000	17,000 25,000 33,000 36,500		13,000 20,000 27,000 30,000 36,000	49,000 15,000 21,500 423,500 27,000		48,000 13,500 19,500 421,500 124,000
		Diso		Creep strength (psi) at 1000 hr for creep rates of							
		(a)			0.0001	parcent/hr			0.00001 p	ercent/hr	
	NB-76B-1	(forged)	·		20,000				13,000		
•	MR-76B-Q (quemche		Q (quemched)		19,	,000			10,	500	

"Heat treatments:

HR-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool. HR-76B-Q: Heat-treated and aged; 2300° F, 2½ hr, water-quenched; 16 hr at 1400° F, air-cool.

bhavy data from oresp tests; 0.505-in,-diameter specimens with a 2-in, gage length.

**TAGA data from stress-rupture tests; 0.250-in,-diameter specimens with a 1-in, gage length,

**Estimated values.





TABLE VI -- DATA FOR STRESS AND TIME FOR TOTAL INSTIBNATION AT 1500° I FOR S-816 ALLOY DIBOS NR-76B Navy data

T			Initial		Time (h	r) for total de	formations of			Transition t	o third-stage creep		
Diso (a)	Specimen number	Stress (psi)	deformation (percent)	0.1 percent	0,2 percent	0.5 percent	1 percent	2 percent	5 percent	Time (hr)	Deformation (percent)		
HR_76B-F (forged)	byly byly byly byly dyllo dyllo dyllo dyllo dyllo dyllo dyllo dyllo dyllo	8,000 10,000 10,000 13,000 13,500 16,500 18,000 21,000 25,000	0,033 ,092 ,044 ,080	235 557 175 8 4 1	900, 92150 155 30 43 2 2	1325 230 230 230 23 24	535 68 51 30	970 145		1140 500 170 100 25	 		
NB-76B-Q (quenched)	poly poly poly poly poly poly poly poly	8,000 10,000 13,003 17,500 19,500 22,000 25,000	,045 ,052 ,072	210 20 20 20 	885 805 15 4	190 96 19 6	510 918 44 17	 806 340 76 30	 	540 130 21 16	1.0 .6 .5		
D1	80	T	otal deformation	n			Stress (psi)	to cause tota	l deformation	in	· · · · · · · · · · · · · · · · · · ·		
(a	.)		(percent)		1 hr	10 hr		100 hr	100	00 hr	2000 hz-		
NR-7 (for			0.1 .2 .5 1.0 Transition		16,200 22,000	18,500 18,000 23,000 925,000 24,000		9,000 13,800 17,000 18,500 18,000	17,000 11,20 18,500 12,00		°4,500 8,500 °9,500 °10,200 °10,300		
NB-7 (quen	rGB—Q uobed)		.1 .2 .5 1.0 Transition		22,000 °28,000	°12,500 18,000 23,500 °25,800 °24,500		21,000		5,500 0,000 8,600 1,700 913,400 1,500 914,500 1,500			
<u> </u>		Disc			Oreop strength (pai) at 1000 hr for oreop rates of								
		(a)			0.0001 percent/hr 0				0,000	0001. percent/hr			
	RB-76	iB-₹ (forge	a)		`11,000					8,500			
	160876	B–Q (queno	nhed)	·	13	,500			<u>-</u>	7,500			

*Heat treatments:

ER-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool.

ER-76B-Q: Heat-treated and aged; 2300° F, 25 hr, water-quenched; 16 hr at 1400° F, air-cool.

blata from oresp tests; 0.505-in,-diameter specimens with a 2-in. gage length.

Office and values.

Clastificated values.

Cl





TABLE VII.- COMPARISON OF ROOM-TENCHEATURE AND MICH-TENCHEATURE PROPERTIES OF REVERAL LARGE PURGED DIECS OF LOW-CARDON N-155, 8-590, AND 8-816 ALLOYS

Test temperature, OF		L		1200						
Allay	Low-carbon #-155 (b)		-590 (a)	B{	11.6	Low-carbon W-155 (b)		-59° (8)	8-	81.6
Disc number ^A	100 - 660	743-7	357-7 ¹ 3-QA	ED-768-F	753-7638-Q	JEB- 66D	100-74B-P	NR-743-QA	103-768-17	338-7638-Q
Short-time properties: Charpy impact strength, foot-pounds	5 22	5	9	25	19	51	9	15	43	43
Ized impact strength, foot-pounds Tamelle strength, psi O.l-percent-offset	56 118,000 69,000	6 129,050 90,000	7 130,500 63,500	18 150,000 79,000	19 144,000 70,000	83,000 47,500	88,700 66,2 5 0	81,600 46,000	180,000 63,000	106,000
yield strength, pai 0.2-percent-offset yield strength, pei	72,500	98,250	70,500	85,0∞	76,000	50,000	71,750	19,000	67,000	58,000
Elongation, percent	35.4	8	17	हा	30	57	15	27	16	12
Empture strengths, pai: 10-hr 100-hr 1000-hr						65,000 55,000 42,000	469,000 52,500 40,000	466,000 72,000 42,000	478,000 62,000 50,000	484,000 66,000 53,000
Oreep strengths, psi: 0,0001 percent/hr 0,00001 percent/hr						26,000 15,000	27,500	23,000	28,000 418,000	98,000 416,000
100-br deformation strengths, pair 0.1-percent deformation 0.5-percent deformation 1.0-percent deformation Transition						17,500 26,000 35,000 40,000 51,500	28,500 36,000 42,000 47,000	26,000 33,800 39,500 49,000	31,500 43,000 48,000 59,500	d19,500 94,500 46,000 92,500 464,000
1000-br deformation strengths, pail 0.1-percent deformation 0.2-percent deformation 0.5-percent deformation 1.0-percent deformation Transition						14,500 24,000 30,000 35,000 39,500	22,000 32,000 34,300 39,000	d18,500 27,000 33,000 39,000	24,500 33,500 58,000 48,000	25,500 37,000 413,000 52,000
							After o	reep testing at	1900° F	
Residual room-temperature properties: Incd impact strength, foot-pounds						17.5			11	5.5
Tensile strongth, pei O.L-parcent-offset rield strongth, pei		~~~~~				118,750 72,500	127,000 85,000	131,000 78,000	139,000 79,000	138,000 81,000
0.2 percent-offset yield strength, pei					- 4-4	76,250	94,500	85,000	87,000	88,000
Elongation, percent						85 `	6	6	8.0	8.5

The treatments:

HM-66D: Forged disc; stress-relieved by besting to 1200° F for 2 hr and cooling in air.

HM-74B-74: As-forged disc; aged 16 hr at 1400° F, air-cool.

HM-74B-94: Heat-treated and aged disc; 2300° F, 3 hr, water-quenched; 16 hr at 1400° F, air-cool.

HB-76B-F: As-forged and aged disc; 16 hr at 1400°F, air-cool.
HB-76B-Q: Esst-treated and aged disc; 2300°F, 25 hr, water-guenched; 16 hr at 1400°F, air-cool.

Duta from reference 3. OData from reference 10. dEstimated values.



THIS DOCUMENT PROVIDED BY THE ABBOTT AEROSPACE TECHNICAL LIBRARY

TABLE VII. - COMPARISON OF HOOM-TEMPERATURE AND HERE-TEMPERATURE PROFESTION - Concluded

Test temperature, °T			1350					1500		· ·
Alloy	Iow-carton W-155 (b)	a_ (-590 o)	8-8	16	Low-carbon W-155 (b)	B- (990 o)	8-	-816
Disc rander	192966ED	101-743- 3 *	103-74 B-QA	113-76B-1	103-768 - -0	NDR-6600	16R-74B-2	MR-74B-QA	ин-76в-г	ш в-76в-q
Short-time properties: Unarpy impact abrength, foot-pounds	50	11	17	47	l _k o	46	13	20	43	43
Isod impact strength, foot-rounds		*								4
Tensile strength, psi 0.1-percent-offset	60,000 42,000	64,625 50,750	65,750 43,500	88,000 56,000	83,000 92,000	40,500 33,000	43,125 31,350	44,400 35,050	59,000 46,000	60,000 49,000
yield strongth, pei 0.2-percent-offset	44,500	55,000	46,000	59,000	55,000	34,000	35,900	37,850	₩9,000	51,000
yield strongth, pai Elongation, percent	24.2	89	25	23	98	30.5	25	18	17	1.7
Rupture strongths, wai: 10-hr 100-hr 1000-hr	\$4,000 \$1,000 \$6,000	442,000 27,500 18,000	⁰ 41,000 32,000 25,000	52,000 37,500 27,000	453,000 39,000 29,000	427,300 20,000 14,200	429,000 13,100 6,000	20,000 15,000	431,000 20,500 13,700	429,500 22,600 17,500
Orosp strengths, psi: 0,0001 percent/hr 0,00001 percent/hr	16,000 7,900	10,600	16,400 12,100	19,500 13,000	19,000 10,500	8,700 65,000	d2,800	10,000 7,100	11,000	13,500 7,500
100-br deformation strengths, pel: 0,1-percent deformation 0.2-percent deformation 0.5-percent deformation 1.0-percent deformation Transition	11,000 16,700 22,000 25,000	13,100 17,000 20,500 24,500	14,700 21,400 24,100 29,000	13,000 28,000 26,000 31,000 35,500	13,000 20,000 27,000 30,000 36,000	7,700 11,000 15,500 17,400 16,400	6,500 9,000 9,300	9,400 11,000 14,800 17,300 16,700	9,000 13,800 17,000 18,500 18,000	9,000 14,000 19,000 21,000 60,200
1000-br deformation strengths, psi: 0.1-percent deformation 0.3-percent deformation 0.5-percent deformation 1.0-percent deformation Transition	8,000 12,000 17,200 19,500 18,000	48,000 13,000 15,500 14,500	8,700 17,000 20,800 22,500	410,000 16,500 24,500 426,500 28,000	49,000 15,000 21,500 423,500 27,000	45,300 6,800 10,500 12,000 11,200	44,000 d4,200	47,300 9,200 11,600 13,600 12,800	85,500 9,600 11,800 12,000 12,000	d5,500 10,000 d1k,700 d16,000 d16,000
D. 12. 1 4		After a	eep testing at	1350° 7			After a	meep testing at	1500° F	•
Residual room-temperature properties: Isod impact strength,	4.6	8	4	7	7.8	4.5		δ	5.5	4.8
foot-pounds Tensile strength, pei 0.1-percent-offest	126,500 65,500	110,500 76,000	132,500 72,000	136,500 82,000	133,500 75,500	114,000 50,500	105,000 71,200	116,000 55,000	123,000 67,000	119,000 65,000
yield strength, pei 0.8-percent-offset	69,500	85,000	79,000	89,000	81,000	54,500	80,800	62,500	75,500	71,500
yield strongth, pei Elongation, percent	13	1	3	9.0	10.7	15	1.5	5	7.4	7.0

*Heat treatments:

HH-66D: Forged disc; stress-relieved by heating to 1200° F for 2 hr and cooling in air.

HH-74B-F: As-forged disc; aged 16 hr at 1400° F, air-cool.

IR-74B-04: East-treated and aged disc; 2500° F, 3 hr, vater-quenched; 16 hr at 1400° F, air-ocol.

BB-76B-F: As-forged and agod disc; 16 hr at 1400° F, air-cool.

HB-76B-Q: Host-treated and aged disc; 25000 F, 2 hr, water-guenohed; 16 hr at 14000 F, air-cool.

Data from reference 3. OData from reference 10. dEstimated values.



TABLE VIII.- OREEP TEST DATA AT 12000, 13500, AND 15000 F FOR 8-816 ALLOY DISCS TR-76B All specimens were 0.505-in. in diameter with a 2-in. gage length

Disc	Specimen	Test temperature		Duration	Initial deformation	Cree	p rate (p	ercent/hr) at -	Total	deformat	ion (per	ent) at -
(a)	number	(°F)	(psi)	(hr)	(percent)	500 hr	1000 hr	1500 hr	2000 hr	500 hr	1000 hr	1500 hr	2000 hr
NR-76B-F (forged)	plesz plesz	1200 1200	25,000 35,000	1124 1124	0.087 .133	0.000083 .000415	0.000050			0,172 ,498	0,205 ,688		
	ofizi ofizi ofizi ofizi	1350 1350 1350 1350	12,000 15,000 20,000 25,000	2016 1344	.057 .072 .080 .094	.000050 .000076 .000172 .000195	.000046			.168	.126 .195 .336 .382	0.137 .218 d.370 .450	0.145 .235 .517
	CF1Y CF13Y CF1Z CF7Z	1500 1500 1500 1500	8,000 10,000 10,000 13,000	2490 1995 2010 1956	.033 .052 .044 .080	.000065 .000105 .000055 .000240	.000045 .000065 .000055	.000022 .000043 .000032 .000250	.000020 .000038 .000017 .000305	.125 .170 .129 .307	.146 .210 .160 .420	.160 .231 .182 .545	.167 .246 .194 .675
NR-76B-Q (quenched)	bq6z bq6y	1200 1200	24,000 35,000	1008 1008	.088 .133	.000065	.000050 .000270			.148 -352	.176 .490		
•	o o ohl oos oos ooh oohl	1350 1350 1350 1350 1350	12,000 12,000 15,000 18,000 21,840	942 2040 2012 2012	.057 .053 .072 .078 .096	.000065 .000030 .000095 .000175 .000305	.000020 .000040	.000022 .000015 .000024 .000062	.000022 .000012 .000030 .000060	.129 .101 .194 .281	.154 .113 .223 .337 f.515	.167 .123 .233 .373	.178 .128 .258 .400
	°Q5Y °Q5X °Q5Z	1500 1500 1500	8,000 10,000 13,000	1963 2137 2002	.045 .052 .072	.000055 .000075 .00019	.000027 .000046 .00012	.000015 .000027 .000094	.000015 .000027 .000094	.118 .182 .262	.141 .208 .332	.151 .224 .383	.157 .243 .433

aHeat treatments:

MB-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool.

NR-76B-Q: Heat-treated and aged; 2300° F, $2\frac{1}{2}$ hr, water-quenched; 16 hr at 1400° F, air-cool.

bMACA data.

CNavy data.

At 1344 hr, when test was discontinued.

Cat 1344 hr, when test was discontinued.

Seconds of controller failures, the temperature was high at 585 hr for 1 hr, maximum 1420° F; and at 650 hr for $3\frac{1}{2}$ hr,

maximum 1470° F. The oreep rates and deformations may have been increased by these temperature rises. Palmes at 942 hr, when test was discontinued.



TABLE IX.- EFFECT OF CREEP TESTING AT 1200°, 1350°, AND 1500° F ON THE BOOM-TEMPERATURE PHYSICAL PROPERTIES OF 8-816 ALLOY DIBOS MB-76B

Disc (a)	Specimen	Prior testing conditions			Residual room-temperature properties									
			Stress	Time (hr)	Tensile strength (pai)	Offset yield strongth (psi)			Proportional	Elongation in 2 in.	Reduction of area	Izod impact strength (ft-lb)		Vickers
		Temperature (OF)	(psi)			0.02 percent	0.1 percent	0.2 percent	(psi)	(percent)	(percent)	Navy	NAGA	hardness
NB-76B-F (forged)	рЪйХ	(c)	(0)	(a)		·							18, 18	592
	(d) (e)	(0)	(c)	(0)	150,000	63,000	79,000	85,600	44,200	20.5	20.5	f17.8		r ₃₂₀
	pL5Z	1200	35,000	1124									10, 12	335
	pkgl	1500	25,000	1124	139,000	62,500	79,000	87,000	45,000	8	8.7			
	d _{FSX}	1350	15,000	2016								7.0, 7.0		334
	d _{F13} 1	1350	12,000	2065	136,500	64,000	82,000	89,000	45,000	9.0	6.2			
	d y13 Y	1500	10,000	1995					,			5.0, 6.0		353
}	dyly	1500	8,000	2490	123,000	53,000	67,000	75,500	36,000	7.4	8,2			
HB-76B-Q (quenched)	, pGLI	(0)	(0)	(0)									10, 10	293
	(d) (e)	(0)	(0)	(o)	144,000	56,500	70,600	76,500	33,300	25,0	21.9	€19.0		B326
	p ⁶²	1200	35,000	1008									6, 5	320
	^b Q6Z	1200	24,000	1008	138,000	62,000	81,000	88,000	30,000	8.5	8.7			
	a _{QQ} Z	1350	12,000	20140								7.0, 8.5		319
	dony	1350	18,000	2120	133,500	63,500	75,500	81,000	48,000	10.7	11.9			
	d _{Q5X}	1500	10,000	2137								3.5, 6.0		312
	₫ _{Q5} Y	1500	8,000	1965	119,000	50,500	65,000	71,500	31,500	7.0	7.9			

^aHeat treatments:

MR-76B-F: As-forged and aged; 16 hr at 1400° F, air-cool.

MR-76B-Q: Heat-treated and aged; 2300° F, 22 hr, water-quenched; 16 hr at 1400° F, air-cool.

dwayy data (0.450-in.-diameter impact specimen with a V-notch).

Tensile data from average of four tests on center- and surface-plane radial specimens at rim of disc.

SImpact value is average of values from three tests (QHY - 14.3, QZZ - 22.8, Q3Z - 20.0); Vickers hardness value is average of several impressions from each of two specimens (Q11Y and a specimen from an unidentified location).

bmaca data (0.365-in.-square impact specimen with a 0.050-in.-deep W-notch).

Original condition.

Impact value is average of values from four tests (F17Z - 21.0, 15.0; F18Z - 14.0, 21.0); Vickers hardness value is average of several impressions from each of two specimens (F17Z, F18Z), taken about 1/2 to 1 in. in from the worked surface of the disc.



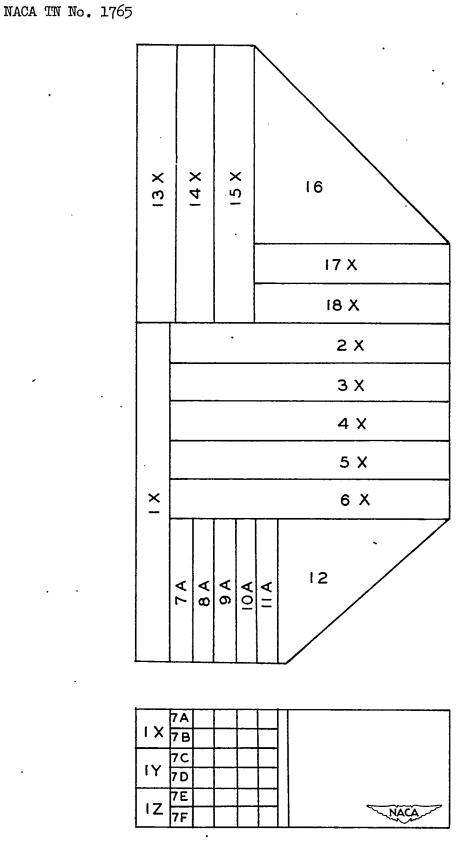


Figure 1.— Location of test coupons from as-forged and aged S-816 alloy disc NR-76B-F. $$



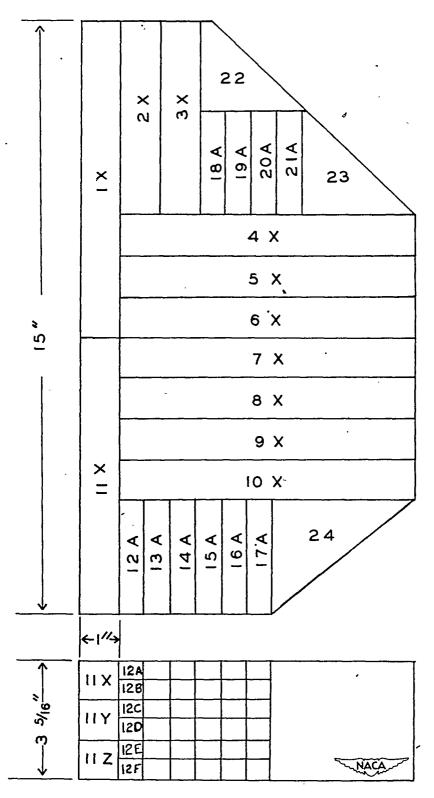


Figure 2.- Location of test coupons from solution—treated and aged S-816 alloy disc NR-76B-Q.

NACA TN No. 1765

TECHNICAL LIBRARY

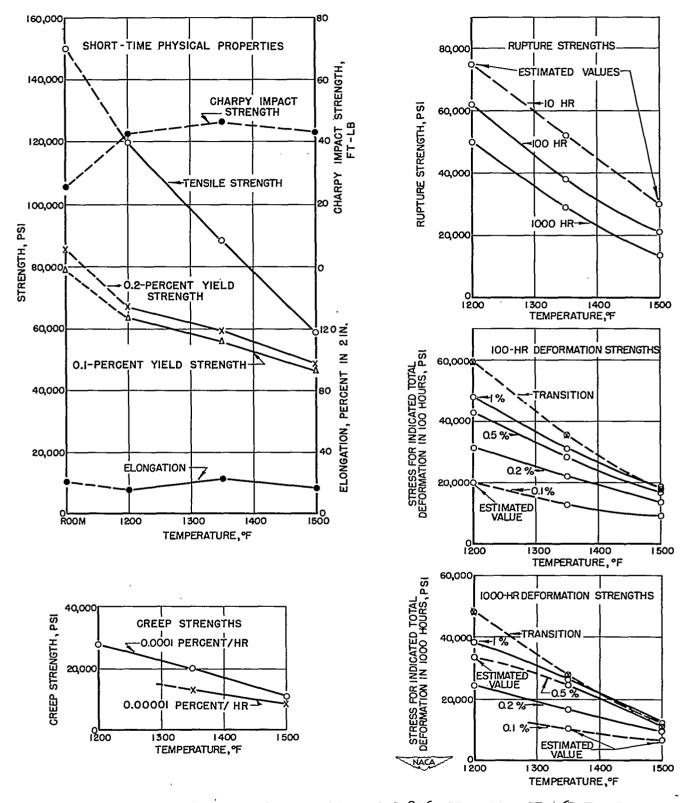


Figure 3.— Summary of properties of S-816 alloy disc NR-76B-F. Disc treatment: As-forged and aged for 16 hours at 1400° F.



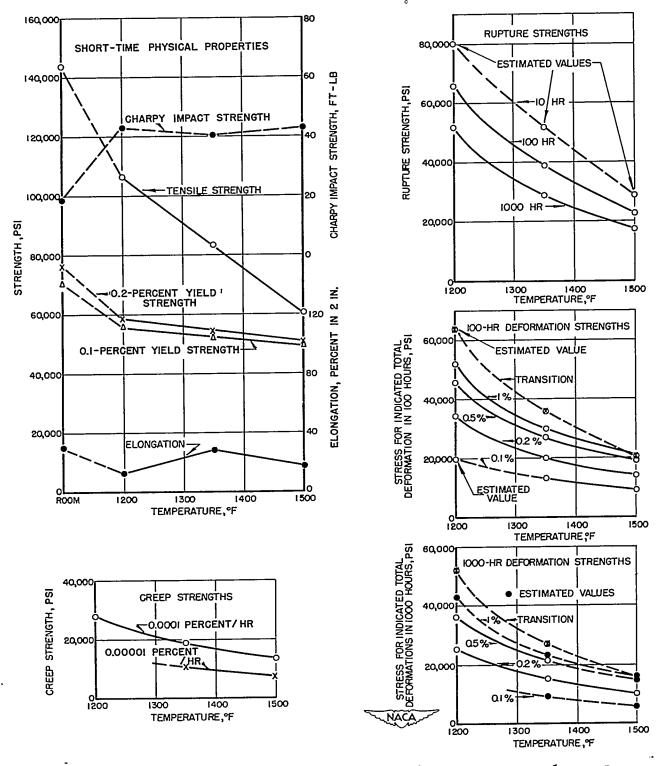


Figure 4.— Summary of properties of S-816 alloy disc NR-76B-Q. Disc treatment: As-forged, water-quenched after $2\frac{1}{2}$ hours at 2300° F, and aged for 16 hours at 1400° F.



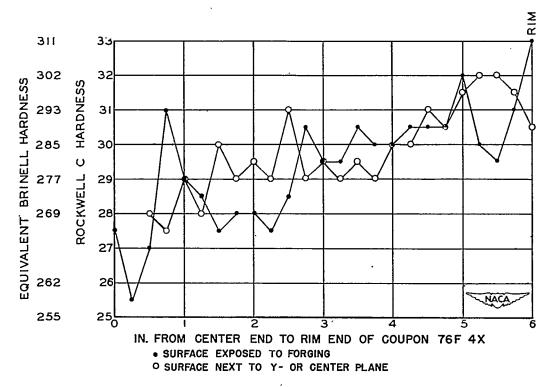


Figure 5.- Variation in hardness from center to rim of as-forged and aged S-816 alloy disc NR-76B-F.

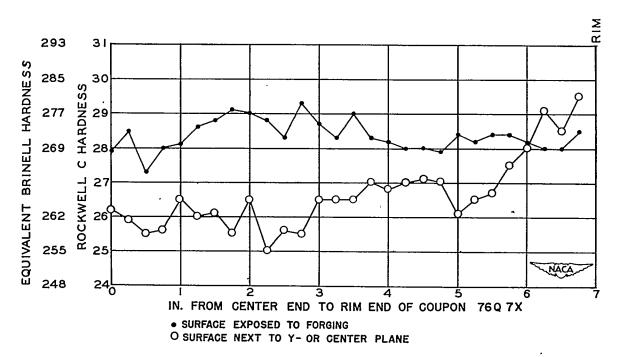


Figure 6.— Variation in hardness from center to rim of solution—treated and aged S-816 disc NR-76B-Q.

•



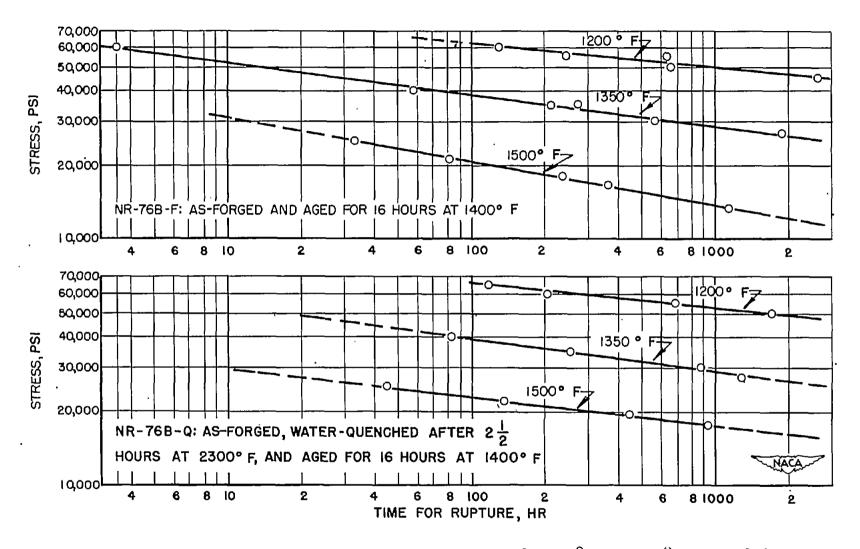


Figure 7.— Curves of stress against rupture time at 1200° , 1350° , and 1500° F for S-816 alloy discs NR-76B.

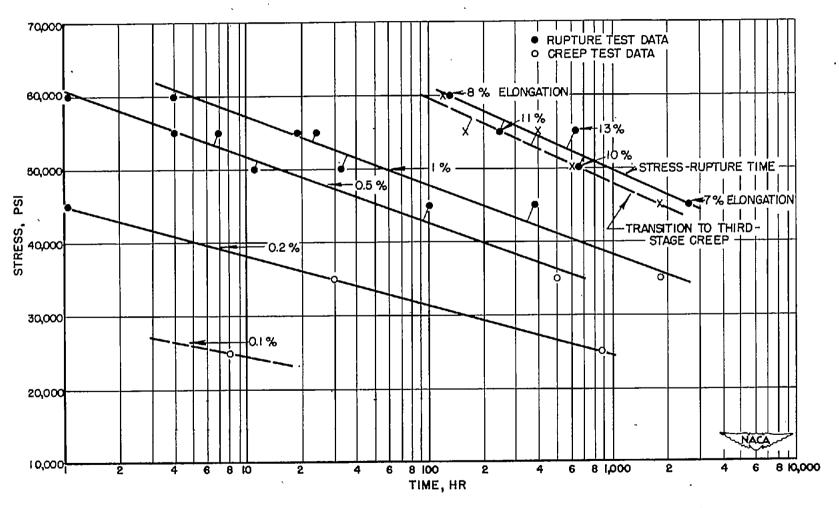


Figure 8.— Curves of stress against time for total deformation at 1200° F for S-816 alloy disc NR-76B-F. Heat treatment: As-forged and aged for 16 hours at 1400° F.



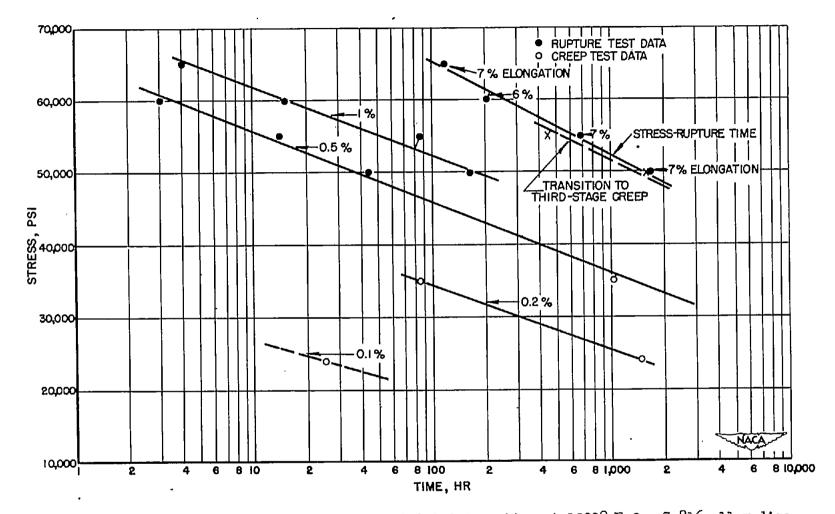


Figure 9.— Curves of stress against time for total deformation at 1200° F for S-816 alloy disc NR-76B-Q. Heat treatment: As-forged, water-quenched after $2\frac{1}{2}$ hours at 2300° F and aged for 16 hours at 1400° F.

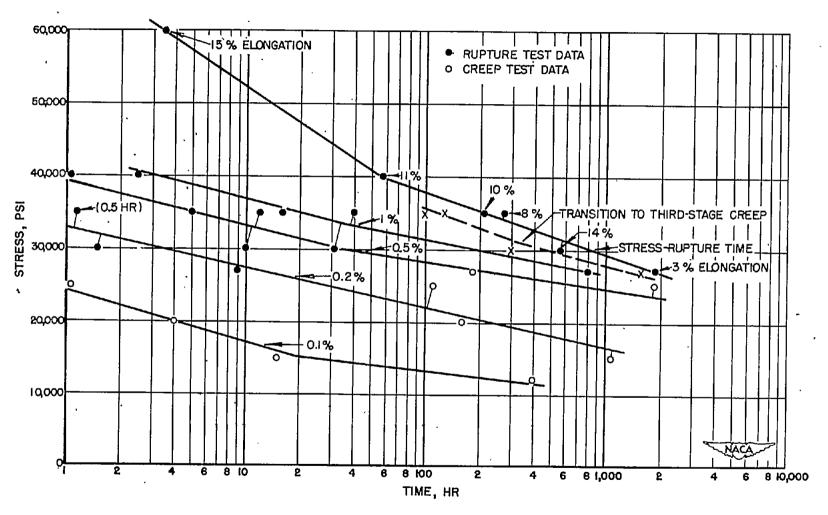


Figure 10.— Curves of stress against time for total deformation at 1350° F for S-816 alloy disc NR-76B-F. Heat treatment: As-forged and aged for 16 hours at 1400° F.

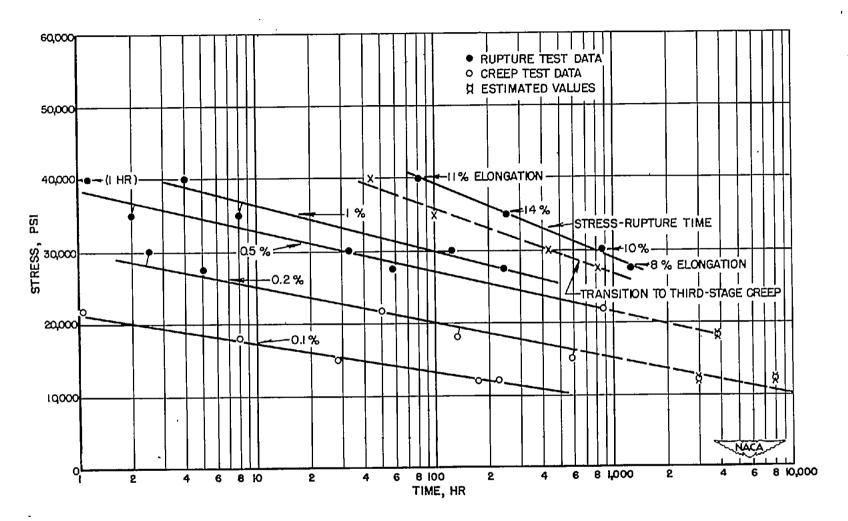


Figure 11.— Curves of stress against time for total deformation at 1350° F for S-816 alloy disc NR-76B-Q. Heat treatment: As-forged, water-quenched after $2\frac{1}{2}$ hours at 2300° F, and aged for 16° hours at 1400° F.

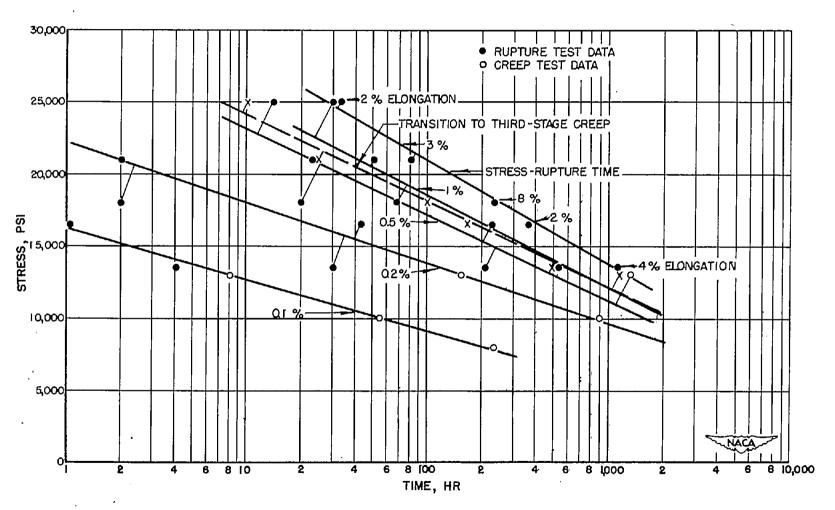


Figure 12.— Curves of stress against time for total deformation at 1500° F for S-816 alloy disc NR-76B-F. Heat treatment: As-forged and aged for 16 hours at 1400° F.

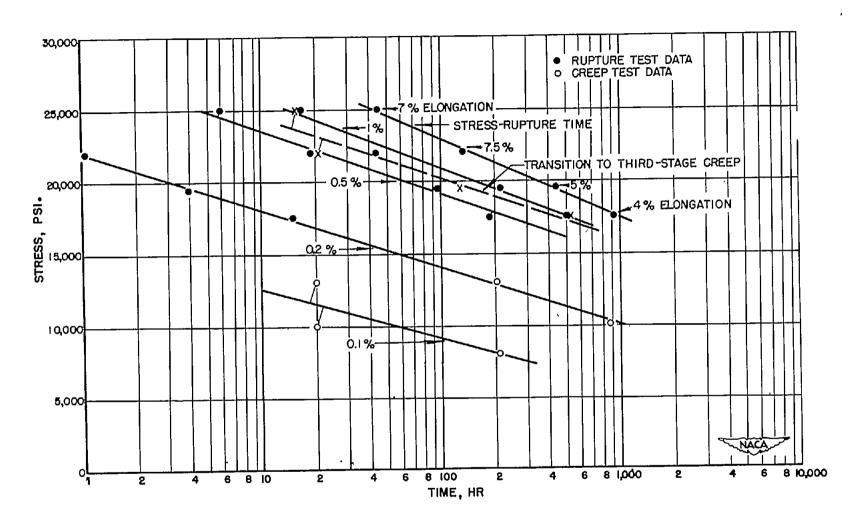


Figure 13.— Curves of stress against time for total deformation at 1500° F for S-816 alloy disc NR-76B-Q. Heat treatment: As-forged, water-quenched after $2\frac{1}{2}$ hours at 2300° F, and aged for 16 hours at 1400° F.

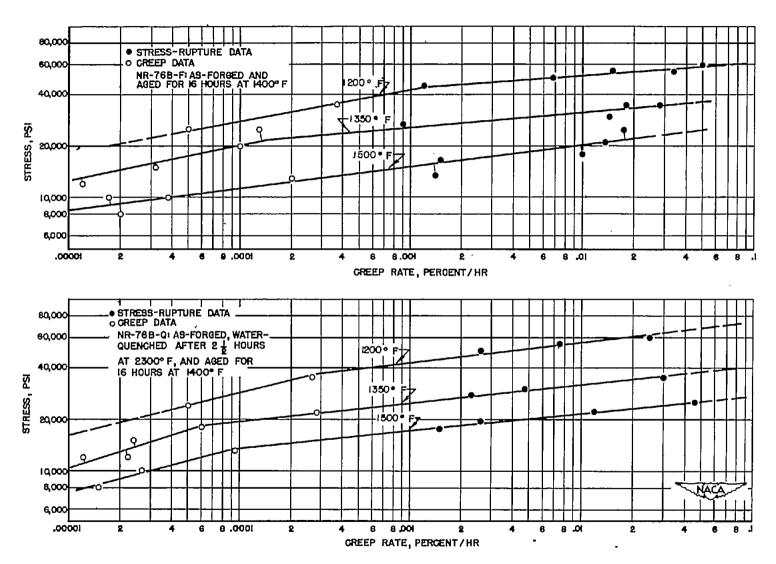


Figure 14.- Curves of stress against creep rate at 1200°, 1350°, and 1500° F for S-816 alloy discs NR-76B.

TECHNICAL LIBRARY

ABBOTTAEROSPACE.COM

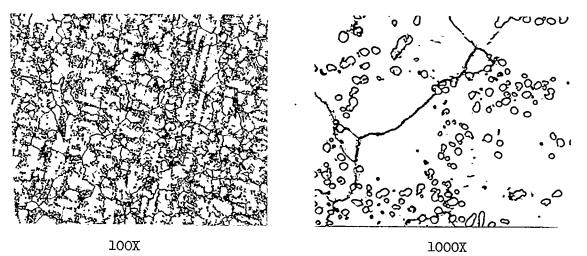




100X 1000X
(a) Disc NR-76B-F; as-forged and aged. Radial section of Y-specimen near rim.



100X 1000X
(b) Disc NR-76B-F; as-forged and aged. Radial section of Y-specimen near center.

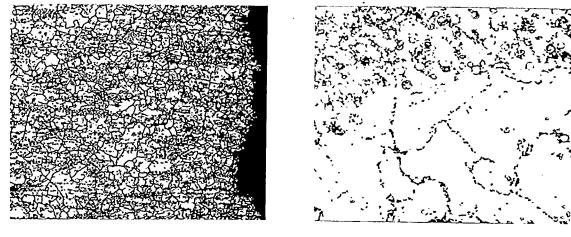


(c) Disc NR-76B-Q; heat-treated and aged. Radial section of Y-specimen near rim.

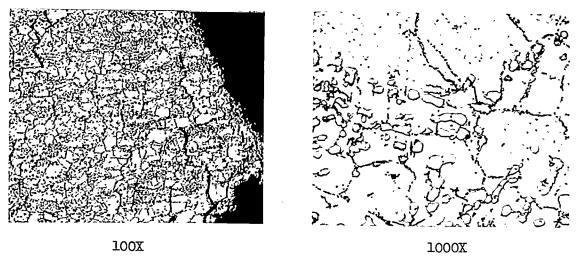
Figure 15.— Original microstructures of S-816 alloy discs NR-76B-F and NR-76B-Q. Electrolytic chromic acid etch. NACA



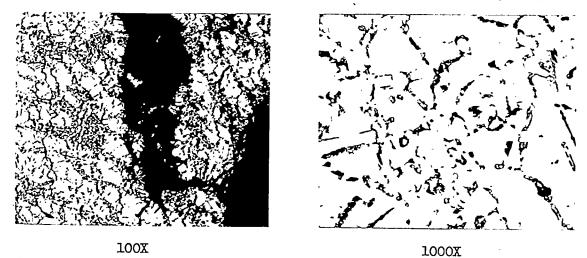




100X 1000X (a) Specimen F9A; 2618 hours for rupture at 1200° F and 45,000 psi.

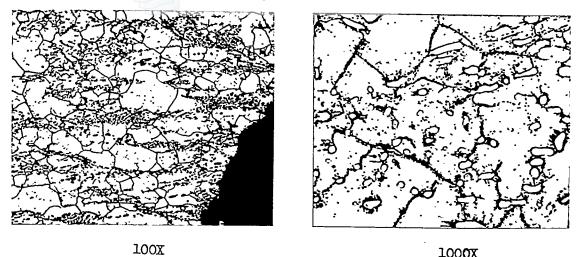


(b) Specimen FlOA; 1894 hours for rupture at 1350° F and 27,000 psi.

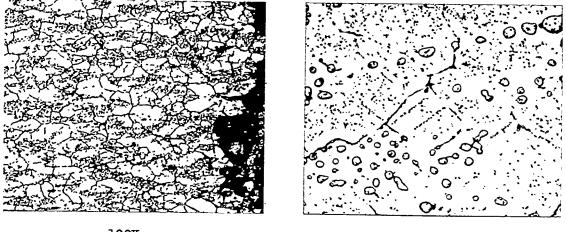


(c) Specimen F11D; 1132 hours for rupture at 1500° F and 13,500 psi. Figure 16.— Microstructures of specimens of S-816 alloy disc NR-76B-F after stress-rupture tests. Disc treatment: As-forged and aged for 16 hours at 1400° F. Electrolytic chromic acid etch.

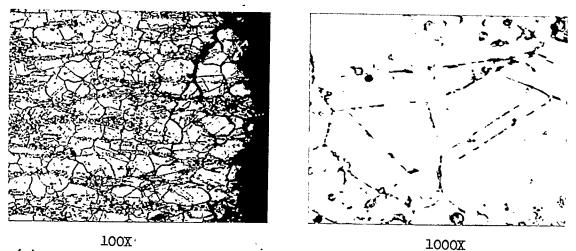




(a) Specimen Q12E; 1699 hours for rupture at 1200° F and 50,000 psi.



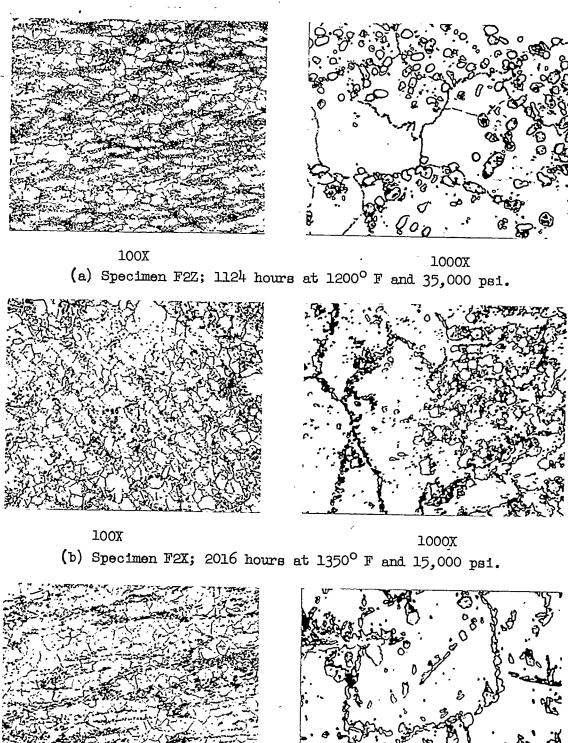
100X 1000X (b) Specimen Q12F; 1286 hours for rupture at 1350° F and 27,500 psi.



(c) Specimen Q14D; 928 hours for rupture at 1500° F and 17,500 psi.

Figure 17.— Microstructures of specimens of S-816 alloy disc NR-76B-Q after stress-rupture tests. Disc treatment: As-forged, water-quenched after 2½ hours at 2300° F, and aged for 16 hours at 1400° F. Electrolytic chromic acid etch.





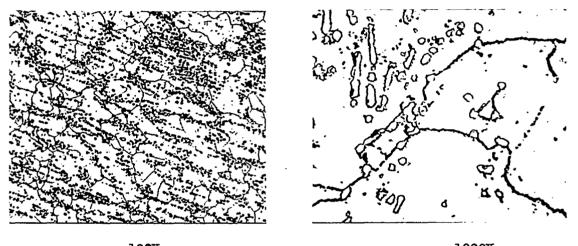
(c) Specimen F13Y; 1995 hours at 1500° F and 10,000 psi.

Figure 18.— Microstructures of specimens of S-816 alloy disc NR-76B-F after creep tests. Disc treatment: As-forged and aged for 16 hours at 1400° F. Electrolytic chromic acid etch.

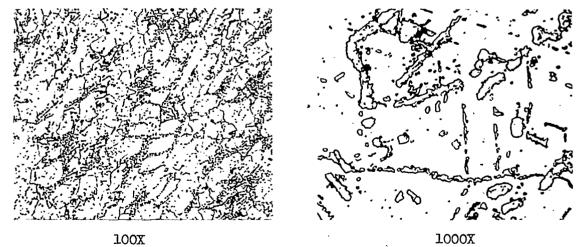




100X 1000X (a) Specimen Q6Y; 1008 hours at 1200° F and 35,000 psi.



100X 1000X (b) Specimen Q9Z; 2040 hours at 1350° F and 12,000 psi.



(c) Specimen Q5X; 2137 hours at 1500° F and 10,000 psi. Figure 19.— Microstructures of specimens of S-816 alloy disc NR-76B-Q after creep tests. Disc treatment: As-forged, water-quenched after $2\frac{1}{2}$ hours at 2300° F, and aged for 16 hours at 1400° F. Electrolytic chromic acid etch.